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Electron spin resonance of diluted solid solutions of Gd_2O_3 in CeO_2

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Abstract

Electron spin resonance spectra of Gd^{3+} in diluted solid solutions of Gd_2O_3 in CeO₂ have been studied at room temperature for Gd concentrations between 0.01 and 1.00 mol%. While in the case of Mn^{2+} :CeO₂ samples, both the linewidth and the line intensity go through a maximum between 0.2 and 0.4% Mn and then start to decrease, in the case of Gd^{3+} :CeO₂ samples the linewidth and the line intensity increase monotonically with the dopant concentration. This as taken as evidence that in Gd_2O_3 –CeO₂ diluted solid solutions there are no clustering effects similar to the ones observed in Mn:CeO₂ solid solutions. It is not clear why clustering effects are present in Mn:CeO₂ solid solutions and not in Gd:CeO₂ solid solutions; however, it seems reasonable to assume that this is due to the fact that the ionic radius of Mn^{2+} (81 pm) is about 25% smaller that that of Gd^{3+} (107.8 pm). In any case, the fact that $Gd:CeO_2$ solid solutions do not exhibit clustering effects means that ESR linewidth data can be used to estimate the concentration of Gd in CeO₂ samples, as it is possible to do in several solid solutions of paramagnetic ions in ceramic materials. The results also suggest that the range of the exchange interaction between Gd³⁺ ions in CeO₂ is about 0.89 nm.

Keywords: Cerium oxide; Gadolinium; Electron spin resonance

1. Introduction

In a recent investigation [1], the study of electron spin resonance spectra of Mn^{2+} showed significant clustering effects in MnO_2 -CeO₂ solid solutions. In the present work, we studied the electron spin resonance spectra of another S-state ion, Gd³⁺, in Ge₂O₃-CeO₂ solid solutions. This investigation is of particular interest because Gd-doped CeO₂ has been proposed [2,3] as a material for intermediate temperature solid oxide fuel cells (IT-SOFCs).

2. Background

2.1. ESR of gadolinium-doped cerium oxide

ESR measurements in single-crystal gadoliniumdoped cerium oxide [4] show that trivalent gadolinium ions substitutionally replace cerium ions the lattice. The spectrum can be fitted to the Hamiltonian

$$\mathscr{H} = g\beta \vec{H}.\vec{S} + B_4 \big[O_4^0 + 5O_4^4 \big] + B_6 \big[O_6^0 - 21O_6^4 \big]$$
(1)

with g = 1.991, $B_4 = 860 \times 10^{-4}$ cm⁻¹ and $B_6 = 2.8$ cm⁻¹. A recent investigation [5] has shown that this spectrum does not change significantly during heat

treatment in gas mixtures such as CO/CO_2 , N_2 and Ar/H_2 , pointing to a very stable charge state.

2.2. ESR of dilute solid solutions

The theory of dipolar broadening in diluted solid solutions was developed in Ref. [6] and extended in

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Refs. [7,8] to take exchange interactions and clustering effects into account. Its main results can be summarized as follows:

- (i) the lineshape is a truncated Lorentzian;
- (ii) the peak-to-peak first derivative linewidth may be expressed as

$$\Delta H_{\rm pp} = \Delta H_{\rm o} + \Delta H_{\rm d} = \Delta H_{\rm o} + c_{\rm l} f_{\rm e}.$$
 (2)

(iii) the intensity of the absorption line is

$$I = c_2 f_{\rm e},\tag{3}$$

where ΔH_0 is the intrinsic linewidth, ΔH_d is the dipolar broadening, c_1 and c_2 are constants and f_e is the concentration of isolated substitutional ions of the paramagnetic impurity, which can be expressed as

$$f_{e} = f(1 - pf)^{z(r_{c})},\tag{4}$$

where *f* is the impurity concentration, $z(r_c)$ the number of cation sites not included in a sphere of radius r_c , r_c the effective range of the exchange interaction and $p \ge 1$ is a "clustering factor" defined [9] as the ratio of the actual probability that a neighbor cationic site is occupied to the probability of occupation if the distribution were random. For a random solution, p = 1; the larger the value of *p*, the stronger are the clustering effects.

3. Experimental procedure and results

3.1. Sample preparation

The samples used in this study were prepared from high-purity CeO₂ and Gd₂O₃ powders by grinding them together and then firing the mixture for 24 h at 1200 °C in air. Actual Gd concentrations were determined by chemical analysis. Room-temperature X-ray diffraction patterns of the samples matched, within experimental error, the spectrum [10] of CeO₂. No other phases were detected.

3.2. Magnetic resonance measurements

All magnetic resonance measurements were performed at room temperature and 9.5 GHz. The spectrum of a typical sample is shown in Fig. 1. All the lines can be identified as turning points of the Hamiltonian given by Eq. (1). In principle, linewidth and intensity data can be extracted from any of the lines in the powder spectrum. We chose the line indicated by an arrow in Fig. 1, that corresponds to the $+3/2 \leftrightarrow +1/2$, $\theta = 45^{\circ}$ transition, because it has a reasonable amplitude and is well separated from the other lines. The results are listed in Table 1.



Fig. 1. ESR spectrum of a CeO_2 sample doped with 0.01 mol% Gd. The arrow shows the line used to measure the linewidth and intensity of the spectrum.

Table 1

Peak-to-peak linewidth $\Delta H_{\rm pp}$ and line intensity $I_{\rm R}$ of the $+3/2 \leftrightarrow$ +1/2, $\theta = 45^{\circ}$ ESR transition of Gd³⁺ in CeO₂ for several Gd concentrations ($T = 300 \,\mathrm{K}$, $\nu = 9.5 \,\mathrm{GHz}$)

f(mol%)	$\Delta H_{\rm pp}$ (mT)	<i>I</i> _R (a.u.)
0.01	0.55	9
0.10	0.97	31
0.20	1.28	54
0.40	1.72	100
0.60	2.00	126
0.80	2.25	146
1.00	2.38	166

4. Discussion and conclusions

The experimental data on the concentration dependence of the peak-to-peak linewidth ΔH_{pp} and the relative intensity $I_{\rm R}$ of the Gd³⁺ spectrum are compared in Figs. 2 and 3 with similar data for Mn²⁺ in the same host lattice, taken from Ref. [1]. While in the case of Mn²⁺:CeO₂ samples both the linewidth and the line intensity go through a maximum between 0.2% and 0.4% Mn and then start to decrease, in the case of



Fig. 2. Concentration dependence of the peak-to-peak linewidth of the spectrum, ΔH_{pp} , in Gd- and Mn-doped CeO₂.



Fig. 3. Concentration dependence of the line intensity of the spectrum, $I_{\rm R}$, in Gd- and Mn-doped CeO₂.

 Gd^{3+} :CeO₂ samples the linewidth and the line intensity increase monotonically with the dopant concentration. This as taken as evidence that in Gd₂O₃-CeO₂ diluted solid solutions there are no clustering effects similar to the ones observed in Mn:CeO₂ [1], Fe:MgO [7,11] and Fe:CaO [12] solid solutions. The theoretical concentration dependence of the peak-to-peak linewidth ΔH_{pp} , given by Eq. (2), is shown in Fig. 4 for p = 1 (no clustering), $\Delta H_0 = 0.55 \,\mathrm{mT}$ and 8 different ranges of the exchange interaction. Since the incorporation of trivalent metal ions in CeO₂ is compensated via oxygen vacancies, the defect structure of CeO_{2-x} should in principle be taken into account in this analysis. However, the Gd concentrations used in this work (between 0.01 and 1.00 mol%) were so low that this effect can be neglected. The values of r_c and $z(r_c)$ for the



Fig. 4. Concentration dependence of the peak-to-peak linewidth, $\Delta H_{\rm pp}$, in Gd-doped CeO₂. The circles are experimental points; the curves represent results of calculations for eight different ranges of the exchange interaction.

Table 2

Values of the effective range of the exchange interaction, r_c , and the number of cation sites not included in a sphere of radius r_c , $z(r_c)$, for the first 8 coordinate spheres of the CeO₂ lattice

n	$r_{\rm c} ({\rm nm})$	$z(r_{\rm c})$
1	0.000	0
2	0.383	12
3	0.541	18
4	0.663	42
5	0.765	54
6	0.886	78
7	0.937	86
8	1.012	134

first 8 coordinate spheres are listed in Table 2, where *n* is the number of the order of each coordinate sphere (n = 1 includes no neighboring sites, and so on). The values of $z(r_c)$ are those appropriate to the fcc lattice of CeO_2 ; the values of r_c were calculated from the lattice constant at room temperature as measured by X-ray diffraction [10], $a_0 = 0.54110$ nm. The experimental data are also shown in Fig. 4. The experimental data fit the theoretical curve n = 6, which corresponds, according to Table 2, to a range $r_c = 0.886 \,\mathrm{nm}$ for the exchange interaction. Fig. 5 shows the theoretical [Eq. (3)] and experimental (Table 1) intensity data. The vertical scale is arbitrary and was chosen so as to provide the best fit of the experimental points to the theoretical curve for p = 1 and n = 6, or $z(r_c) = 78$. The agreement was found to be good.

In Fig. 6 we compare the concentration dependence of the dipolar broadening ΔH_d of the Gd³⁺ spectrum in



Fig. 5. Concentration dependence of the line intensity of the spectrum, $I_{\rm R}$, in Gd-doped CeO₂. The circles are experimental points; the curves represent results of calculations for eight different ranges of the exchange interaction.



Fig. 6. Concentration dependence of the dipolar broadening of the spectra of Gd^{3+} in MgO, CaO, SrO and CeO₂. The curves are theoretical: $\Delta H_d = c_1 f (1-f)^{z(r_c)}$ with $c_1 = 4.5$, $z(r_c) = 42$ (MgO), $c_1 = 165$, $z(r_c) = 86$ (CaO), $c_1 = 280$, $z(r_c) = 86$ (SrO) and $c_1 = 400$, $z(r_c) = 78$ (CeO₂). MgO data are from [12], CaO data from [13] and SrO data from [14].

CeO₂ and in other three cubic host lattices, MgO [13], CaO [14] and SrO [15]. The faster increase of the linewidth with Gd concentration in CeO₂ is not consistent with the fact that the lattice constant a_0 in CeO₂, 0.541 nm [10], is larger than in the other three compounds. This is probably due to the fact that random lattice stresses broaden non-Kramers transitions such as the one investigated in the present work, as has been already observed in oxides such as MgO [16] and Al₂O₃ [17,18]. It is not clear why clustering effects are present in $Mn:CeO_2$ solid solutions and not in $Gd:CeO_2$ solid solutions. However, it seems reasonable to assume that this is due to the fact that the ionic radius of Mn^{2+} (81 pm) is about 25% smaller that that of Gd^{3+} (107.8 pm) [19]. In any case, the fact that $Gd:CeO_2$ solid solutions do not exhibit clustering effects means that ESR linewidth data can be used to estimate the concentration of Gd in CeO₂ samples, as it is possible to do in several solid solutions of paramagnetic ions in ceramic materials [20–23].

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